A QMC-Deep Learning Method for Diffusivity Estimation in Random Domains

苏州大学 吕力遥,张智文,陈景润

(1. 苏州大学数学科学学院、苏州大学敬文书院, 江苏省 苏州市 邮编 205001;

2. 香港大学数学系, 中国香港, 邮编 999077;

3. 苏州大学数学科学学院、苏州大学数学交叉研究中心, 江苏省 苏州市 邮编 205001)

(Numerical Mathematics: Theory, Methods and Applications, Vol. 13, No. 4,

pp. 908-927)

陈景润教授

中文摘要:激子扩散在许多有机半导体光电器件的功能中起着至关重要的作用,其中精确的描述需要异质结的精准种植。因为高维随机空间中异质结的参数化远远超出了经典仿真工具的功能。在这里,我们开发了一种基于准蒙特卡洛采样生成训练数据集,并使用深度神经网络提取激子扩散长度对表面粗糙度的函数,其中高精度和空前的效率,产生大量的信息在整个参数空间上。我们的方法提供了一种新的策略来分析界面对激子扩散的影响,并有望通过量身定制的光电功能来辅助实验设计。

英文摘要: Exciton diffusion plays a vital role in the function of many organic semiconducting opto-electronic devices, where an accurate description requires precise control of heterojunctions. This poses a challenging problem because the parameterization of heterojunctions in high-dimensional random space is far beyond the capability of classical simulation tools. Here, we develop a novel method based on Quasi-Monte Carlo sampling to generate the training data set and deep neural network to extract a function for exciton diffusion length on surface roughness with high accuracy and unprecedented efficiency, yielding an abundance of information over the entire parameter space. Our method provides a new strategy to analyze the impact of interfacial ordering on exciton diffusion and is expected to assist experimental design with tailored opto-electronic functionalities.

作者简介: 吕力遥 (1998-)、男, 江苏, 苏州大学数学科学学院数学与应用数学 (基地), 苏州大学敬文书院, 本科四年级, 主要从事计算数学等研究。

张智文(1982-)、男,香港大学数学系助理教授,主要从事偏微分方程数值解和不确定性量化等研究。 陈景润(1982-)、男,苏州大学数学科学学院教授,主要从事材料性质的建模、分析与算法研究。

关键词: Exciton diffusion length; deep learning; Quasi-Monte Carlo sampling, diffusion equation

Introduction

Over the past decades, much attention has been paid on organic semiconductors for applications in various opto-electronic devices^{[1][2][3][4]}. These materials include small molecules^{[5][6]}, oligomers^{[7][8]} and polymers^{[9][10]}. Exciton diffusion is one of the key processes behind the operation of organic opto-electronic devices^{[11][12][13]}. From a microscopic perspective, exciton, a bound electron-hole pair, is the elementary excitation in opto-electronic devices such as light emitting diodes and organic solar cells. The exciton diffusion length (EDL) is the characteristic distance that excitons are able to travel during their lifetime ^[6]. A short diffusion length in organic photovoltaics limits the dissociation of excitons into free charge ^{[14][15]}. Conversely, a large diffusion length in organic light emitting diodes may limit luminous efficiency if excitons diffuse to non-radiative quenching sites^[16].

As quasi-particles with no net charge, excitons are difficult to probe directly by electrical means ^[17]. This is particularly true in organic semiconductors where the exciton binding energy is \sim 1 electronvolt^[18]Reported techniques to measure EDL include photoluminescence (PL) surface quenching^{[6][14][19][20][21][22][23][24][25]}, time-resolved PL bulk quenching modeled with a Monte Carlo (MC) simulation^{[10][26]}, exciton-exciton annihilation^{[27][28][29][30]}, modeling of solar cell photocurrent spectrum^{[31][32][33][34][35][36][37][38][39]}, time-resolved microwave conductance^{[40][41]}, spectrally resolved PL quenching^{[42][43][44]} and F ⁰ rster resonance energy transfer theory^{[42][45][46]}. From a theoretical perspective, the minimal modeling error is given by the diffusion equation model ^[47], which is employed in the current work.

To be precise, the device used in PL surface quenching experiment includes two layers of organic materials with thickness ranging from dozens of nanometers to hundreds of nanometers. One layer of material is called donor and the other is called acceptor or quencher according to the difference of their chemical properties. Under the illumination of solar lights, excitons are generated in the donor layer and diffuse in the donor. Due to the exciton-environment interaction, some excitions die out and emit photons which lead to the PL. The donor-acceptor interface serves as the absorbing boundary while other boundaries serve as reflecting boundaries due to the tailored properties. Since the donor-acceptor interface is not exposed to the air/vacuum and the resolution of the surface morphology is limited by the resolution of atomic force microscopy, the interface is subject to an uncertainty. It is found that the fitted EDL is sensitive to the uncertainty in some scenarios. From a numerical perspective, the random interface requires a parametrization in high-dimensional random space, which is prohibitively expensive for any simulation tool. For example, MC method overcomes the curse of dimensionality but has very low accuracy^[48]. Stochastic collection method has high accuracy but is only affordable in low dimensional random space^[49]. Asymptotics-based method is efficient but its accuracy relies heavily on the magnitude of randomness^{[50].} In the current work, we propose a novel method based on deep learning with high accuracy and unprecedented efficiency.

Recently, increasing attentions have been paid to apply machine learning (ML) techniques to materials-related problems. For example, the classification of crystal structures of transition metal phosphide via support vector machine^[51] leads to the discovery of a novel phase^[52]. Likewise, a hybrid probabilistic model based on high-throughput first-principle computation and ML was developed to identify stable novel compositions and their crystal structures^[53]. Physical parameters such as band gap^{[54][55]}, elastic constants^{[54][56]}, and Debye temperature^[54] have also been predicted using an array of ML techniques. In another line, deep learning (DL) in computer science has had great success in text classification^[57], computer vision^[58], natural language processing^[59], and other data-driven applications^{[60][61]}. Recently, DNNs have been applied to the field of numerical analysis and scientific computing, like inverse problems ^[62] and dimensional partial differential equations (PDEs)^{[63][64][65][66]} One significant advantage of DL is its strong ability to approximate a complex function in high dimensions and extract features with high precision using composition of simple nonlinear units.

Meanwhile, benefiting from recent advances in parallel graphics processing unit - accelerated computing, huge volumes of data can be put into the DL architecture for training.

In this work, we employ DL to extract a complex function of EDL in terms of the random interface parametrized in a high-dimensional space. The fitted function has rich information, which explains a few interesting experimental observations. Compared to classical simulation tools, our approach has the following features: quasi-Monte Carlo (QMC) sampling^{[67][68][69]} for data collection and ResNet^[70] for training. The size of data in the former step grows only linearly with respect reto the dimension of random space, thus our approach overcomes the curse of dimensionality (with possibly a logarithmic growing factor depending on the dimension). With the usage of ResNet in the latter step, a complex function can be extracted with high accuracy.

Our main contribution is the use of QMC sampling to explore the high-dimensional space of surface roughness, i.e., generating realizations of surface roughness by the QMC method; see the left column in Fig. 1. Once a surface roughness is generated, an inverse problem is solved to produce EDL. DL is then used to construct a map between surface roughness and EDL; see the center column in Fig. 1. Afterwards, the full landscape of EDL on surface roughness can be generated and further analyzed; see the right column in Fig. 1. Generally speaking, QMC method has an approximation accuracy independent of dimension which scales inversely with respect to the number of samples (with possibly a logarithmic factor depending on dimension). This significantly outperforms Monte-Carlo method. Other sampling strategies, such as Latin Hypercube Sampling, Hammersley Sequence Sampling, and Latin hypercube-Hammersley sequence sampling, shall also perform better. ResNet is used to construct the map between surface roughness and EDL. Note that the data generation dominates the whole simulation since each data requires solves an inverse problem involving solving an elliptic problem over a curved domain. Therefore, even though there may be better network structures in the sense of accuracy or efficiency, we stress that the one used in this paper has already provided a great choice and only linear growth of number of samples with respect to dimension is needed for accurate training. No

significant change is found if more layers or parameters are added in the ResNet structure we have used.

The rest of this paper is organized as follows. In Section 2, we introduce models for exciton diffusion in 1D, 2D, and 3D, respectively. In Section 3, we propose a QMC-machine learning method based on QMC sampling and ResNet. Using a series of numerical simulations in Section 4, we show that the QMC-machine learning method is both accurate and efficient. Moreover, modeling error and mode dependence can be extracted from the fitted EDL, which explains a few interesting experimental observations. Conclusions are drawn in Section 5.

Model description

(-) 1D model

An exciton that diffuse in the donor follows a diffusion-type equation over a 1D random domain $D_1 = \{x: x \in [\theta(\omega), d]\}, \#(1)$

where the interface is reduced to a random point $x = \theta(\omega)$.

The corresponding diffusion equation is

$$\begin{cases} \sigma^2 u_{xx} - u + G(x) = 0, \theta(\omega) < x < d \\ u_x(d) = 0 & \#(2) \\ u(\theta(\omega)) = 0 \end{cases}$$

where σ is the EDL to be extracted, u is the exciton density, G is the normalized exciton generation function by the transfer matrix method^[71]. x = d serves as the reflecting boundary and Neumann boundary condition is imposed on the boundary exposed in air. $x = \theta(\omega)$ serves as the absorbing boundary and homogenous Dirichlet boundary condition is imposed on the donor-acceptor interface and θ is the input of the network in the 1D case. The photoluminescence (PL) is where the integral is approximated by the Simpson's rule in the simulation.

The 1D model is commonly used to extract the exciton diffusion length (EDL) due to its simplicity and model accuracy^{[6][47]}. However, the 1D model does not always work well^{[6][50]} and we introduce 2D and 3D models for the same problem. They are more realistic models for exciton diffusion. We shall demonstrate their differences in Section 4.

(=) 2D model

The 2D model is defined over a random domain

$$D_2 = \{(x,y) | h(y,\omega) < x < d, 0 < y < L_y\}, #(4)$$

where the interface is a random line parametrized by

where \hat{h} is the magnitude of length due to the roughness limited by the resolution of atomic force microscopy, $\theta_k(\omega)$ are i.i.d. random variables, $\phi_k(y) = \sin\left(2k\pi \frac{y}{L_{\omega}}\right)$ and $\beta < 0$ controls the decay rate of spatial modes $\phi_k(y)$. Note that $\theta_k(\omega)$ are the input of the network in the 2D case. The rougher the interface is, the closer the β approaches 0. Note that the dimensionless magnitude of randomness/perturbation is defined as $\varepsilon = \frac{\hat{h}}{d}$, which is used in asymptotic-based approaches^{[50][72]}. Given the surface roughness measured in an experiment, parameters in (5) can be extracted via discrete Fourier transform.

The corresponding diffusion equation is

$$\begin{cases} \sigma^2 \bigtriangleup u - u + G(x,y) = 0, x \in D_2, \\ u_x(d) = 0 \\ u(h(y,\omega),y) = 0, 0 < y < L_y, \\ u(x,y) = u(x,y + L_y), h(y,\omega) < x < d, \end{cases}$$
(6)

and the PL is

$$I_{\theta(\omega)}[\sigma,d] = \frac{1}{L_y} \int_0^{L_y} \int_{h(y,\omega)}^d u(x,y) dx dy.\#(7)$$

Note that $\overline{L_{w}}$ in (7) is used for the consideration of modeling error between 1D and 2D

models ^[50].

(Ξ) 3D model

The 3D random domain can be defined as

$$D_3 = \{(x, y, z) | h(y, z, \omega_1, \omega_2) < x < d, 0 < y < L_y, 0 < z < L_z\}. #(8)$$

Here the donor-acceptor interface x = n(y, z)

 $x = h(y,z,\omega_1,\omega_2)$ is parameterized by

$$h(y,z,\omega_1,\omega_2) = \hat{h} \sum_{k_1=1}^{K_1} \sum_{k_2=1}^{K_2} k_1^{\beta} k_2^{\beta} \theta_{k_1}(\omega_1) \theta_{k_2}(\omega_2) \phi_{k_1}(y) \phi_{k_2}(z), #(9)$$

where

 $\hat{\mathbf{h}}$ is the magnitude of length due to the roughness limited by the resolution of atomic

force microscopy, $\theta_{k_1}(\omega_1), \theta_{k_2}(\omega_2) \quad \text{are i.i.d. random variables,} \qquad \varphi_{k_1}(y) = \sin\left(2k_1\pi \frac{y}{L_y}\right) \quad ,$

measured in an experiment, parameters in (2) can be extracted via discrete Fourier transform.

The 3D diffusion equation reads as

$$\begin{aligned} \sigma^2 & \bigtriangleup u - u + G(x, y, z) = 0, x \in D, \\ u_x(d, y, z) &= 0, 0 < y < L_{y'}, 0 < z < L_z, \\ u(h(y, z, \omega_1, \omega_2), y, z) &= 0, 0 < y < L_{y'}, 0 < z < L_z, \\ u(x, y, z) &= u(x, y + L_y, z) = u(x, y, z + L_z), h(y, z, \omega_1, \omega_2) < x < d \end{aligned}$$

$$(10)$$

The PL is computed by

$$l_{\theta(\omega_1),\theta(\omega_2)}[\sigma,d] = \frac{1}{L_z} \frac{1}{L_y} \int_0^{L_z} \int_0^{L_y} \int_h^d u(x,y,z) dx dy dz.\#(11)$$

At the formal level, when $L_z \rightarrow 0$, the PL of 3D model defined by (11) reduces to the PL of 2D model defined by (7), and further they reduce to the PL of 1D model defined by (3) as

$$L_y \rightarrow 0$$

.Again, Note that $\frac{1}{L_x} \frac{1}{L_y}$ in (11) is used for the consideration of modeling error

between 1D and 3D models.

In the experiment, PL data $\{\hat{l}_i\}_{i=1}^N$ are measured by a series of bilayer devices with different thicknesses $\{d_i\}_{i=1}^N$, where d_i is the thickness of the *i*-th donor layer. Since there are

not enough experimental data, we generate the reference PL data by solving Equation (2) with a prescribed σ and without surface roughness.

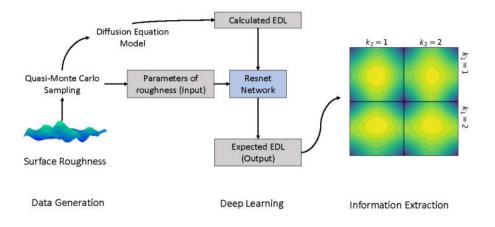
(四) Newton's method for the inverse problem

The optimal EDL σ is expected to reproduce the experimental date $\{d_i, \bar{I}_i\}_{i=1}^N$ in the

sense of minimized mean square error (MSE)

$$\min_{\sigma} J_{\theta(\omega_1),\theta(\omega_2)}(\sigma) = \frac{1}{N} \sum_{i=1}^{N} \left(I_{\theta(\omega_1),\theta(\omega_2)}(\sigma,d_i) - \widehat{I}_i \right)^2, \#(12)$$

where notations are based on 3D model.



Newton's method is used to solve (12) for σ .

Given $\sigma^{(0)}$, for $n = 1, 2 \cdots$, until convergence, Newton's method for (12) solves

$$\sigma^{(n)} = \sigma^{(n-1)} - \alpha_n \frac{\frac{\partial}{\partial \sigma} J(\sigma^{(n-1)})}{\frac{\partial^2}{\partial^2 \sigma} J(\sigma^{(n-1)})}$$

with $\alpha_n \in (0,1]$ given by line search^[73].

The calculated σ is defined as $\sigma_{\theta(\omega_1),\theta(\omega_2)}$. Therefore, for different parameters Figure 1: Flow chart of the deep learning method for extracting exciton diffusion length over the parameter space. Left: data generation; Middle: data training; Right: data prediction. In the stage of data generation, quasi-Monte Carlo method is used to sample the random space, and the actual exciton diffusion length is generated by solving the diffusion equation model. In the stage of data training, a

complex function $\sigma(\theta(\omega_1), \theta(\omega_2))$ is approximated over the entire parameter space. In the stage of

data prediction, given the full landscape of $\sigma(\theta(\omega_1), \theta(\omega_2))$, both qualitative and quantitative information can be extracted.

$$\theta(\omega_1), \theta(\omega_2)$$
, we get a data set $\left\{ \left(\theta(\omega_1)[j], \theta(\omega_2)[j], \sigma_{\theta(\omega_1)[j], \theta(\omega_2)[j]} \right)_{i=1}^M \right\}$ with M the

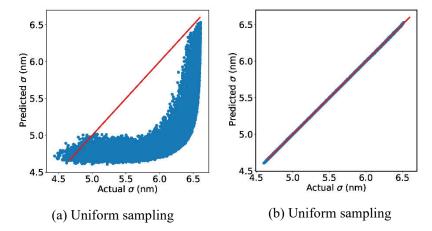
size of data set.

A QMC-machine learning method

The main difficulty of 2D and 3D models in random domains is the high dimension of random variables, and thus it is very difficult to solve these models with accurate results using classical simulation tools, such as MC method, stochastic collocation method, and asymptotics-based methods. Therefore, we propose a QMC-machine learning method to overcome the curse of dimensionality. Our approach consists of four major components: QMC sampling over the high-dimensional random space; diffusion equation model for data generation; ResNet for training to approximate a complex function of EDL; Information extraction for analysis. The flow chart of this process can be seen in Fig. 1. The models are introduced in Section 2 and information extraction for analysis will be discussed in Section 4. For completeness, we introduce QMC sampling and ResNet in this section.

(-) Quasi-Monte Carlo sampling

^M is needed to ensure that the extracted In the sampling stage of data preparation, a large function of EDL has the desired accuracy. There are two classical choices: uniform sampling and ^M grows exponentially fast with respect to K_1 random sampling. For uniform sampling, $K_1 = K_2 = 5$ and points are uniformly distributed K_2 . For example, in the 3D case, if and for each random variable, the size of training data set is shown in Table 1. For the simulations in our work, at least three orders of magnitude reduction in the size of data set is found for OMC sampling strategy. Fig. 2(a) and Fig. 2(b) show the huge advantage of QMC sampling over uniform sampling. 30.561% and L^{∞} error is 0.237% For the same size of training data set, the relative



implying more than two orders of magnitude improvement in the prediction accuracy.

Figure 2: Generalization error of the trained neural network in 2D when the number of random variables is 10 and 1024 points are chosen. The relative L^{∞} error is 30.561% when uniformly distributed points are used and the relative L^{∞} error is 0.237% when points generated by Sobol sequence are used. The training data is visualized as a line and the testing data is visualized as scattering points.

On the other hand, if random sampling is used, then we do not have this issue. However, MC method has poor accuracy $\sim O\left(\frac{1}{\sqrt{M}}\right)$. At least millions of data are needed for training. Meanwhile, for each datum, an inverse problem with the diffusion equation model over a curved domain in 3D has to be solved. These together make the network training prohibitively expensive. Fortunately, compared to uniform sampling and MC sampling, QMC sampling provides the best compromise between accuracy and efficiency. It overcomes the curse of dimensionality and has high accuracy ^{[67][68][69]}. Specifically, we use Sobol sequence to generate points over the (high-dimensional) random space. Fig. 3(b) plots the points generated by Sobol sequence, which is a deterministic way to generate points with better approximation accuracy. QMC sampling has accuracy $\sim O\left(\frac{(\log M)^{dm}}{M}\right)^{[67][68][69]}$, which reduces the size of training data set by orders of magnitudes in comparison with MC method. The size of data in QMC method grows merely linearly fast with respect to the number of random variables.

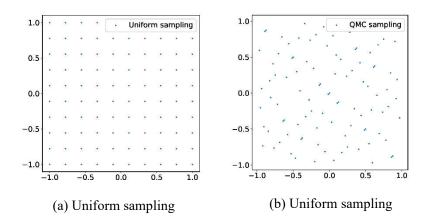


Figure 3: Comparison of two sampling method. (a) Uniform sampling for two random variables with 100 points; (b) Quasi-Monte Carlo sampling (Sobol sequence) for two random variables with 100 points.

(ニ) ResNet

ResNet^[70] is used to approximate $\sigma_{\theta(\omega_1),\theta(\omega_2)}$. A ResNet consists of a series of blocks. One block is given in Fig. 1 with two linear transformations, two activation functions, and one short cut. The ⁱ-th block can be expressed as $t = f_i(s) = g(W_{i,2} \cdot g(W_{i,1} \cdot s + b_{i,1}) + b_{i,2}) + s.\#(13)$

Here $s,t \in \mathbb{R}^m$ are input and output of the ⁱ-th block, and weights $W_{i,j} \in \mathbb{R}^{m \times m}, b_{i,2} \in \mathbb{R}^m$. Sigmoid function, which can be expressed in $g(x) = \frac{1}{1 + \exp(-x)},$

is chosen as the activation function to balance training complexity and accuracy.

The last term in (13) is called the shortcut connection or the residual connection. Advantages of using it are : (1) It can solve the notorious problem of vanishing/exploding gradients automatically. The vanishing gradient problem means that in some cases the gradient will be vanishingly small, effectively preventing the weight from changing its value. In the worst case, this may completely stop the neural network from further training. Also, when activation functions are used whose derivatives can take on larger values, one risks encountering the related exploding gradient problem. (2) Without adding any parameters or computational complexity, the shortcut connection performing as an

mapping can resolve the degradation issue (with the network depth increasing, accuracy gets saturated and then degrades rapidly).

The fully ⁿ-layer network can be expressed as

$$f_w(x) = f_n \circ f_{n-1} \cdots \circ f_1(x)$$

Х ^W denotes the set of parameters in the whole network. Note that the input where R^{\dim} $\sigma(\theta(\omega_1), \theta(\omega_2))_{is in}$ and the output of the whole structure in the first layer is in R^1 . To deal with the problem, we apply two linear transformations on both ^X before putting it into the ResNet structure and on the output of the ResNet structure. For example, we choose $\theta(\omega_2)$ have m = 30, n = 6 $\theta(\omega_1)$ and 5 random in the 3D model. Both $\dim = 10$. Therefore, we apply two linear transforms: one from a 10 variables, and thus 30 dimensional vector and the other from a 30 dimensional dimensional vector to a ¹ dimensional vector before and after the ResNet structure. Parameters in these linear vector to transforms also need to be trained.

The loss function we use is the MSE between the actual EDL $\sigma_{\theta(\omega_1),\theta(\omega_2)}$ given by the diffusion equation model and the predicted EDL $\tilde{\sigma}(\theta(\omega_1)[j],\theta(\omega_2)[j])$ given by the ResNet

$$MSE = \frac{1}{M} \sum_{j=1}^{M} \left(\sigma_{\theta(\omega_1)[j], \theta(\omega_2)[j]} - \tilde{\sigma}(\theta(\omega_1)[j], \theta(\omega_2)[j]) \right)^2, \#(14)$$

where θ represents the parameter set in the ResNet, \mathbf{I} is the \mathbf{L} th sample, and M is the size of training data set.

Define the relative L^{∞} error of EDL as

$$\operatorname{Error} = \max_{1 \leq j \leq M} \frac{|\sigma_{\theta(\omega_1)[j]}|}{|\sigma_{\theta(\omega_1)[j]}|}$$

which will be used to quantify the approximation accuracy of DL.

Result and Discussion

(-) Accuracy and discussion

First, we focus on the 2D problem with only one realization, i.e., only one d = 10 and N = 1. PL data are generated when $\sigma = 10$ without any randomness. Accuracy of the trained neural network in terms of size of the training set is recorded in Table 2.From the results, we can find that a random field with the slower decay rate ($\beta = 0$) is more difficult to be trained when uniform sampling is used.

In the literature, a asymptotics-based method has been proposed^[50], which only works well for random interfaces with small magnitudes. The proposed method works for random interfaces with large $\theta(\omega)$ with 2 random variables ranging over magnitudes. For example, consider [-5,5] and 0.5 $\beta = 2$, i.e., the dimensionless magnitude of perturbation is about therefore the asymptotics-based method has poor accuracy in this case unless enough terms are used in the asymptotic expansion. For example, we use the asymptotics-based method with up to the second 25% However, for the OMC-machine learning order terms^{[50][72]} and the average error is over L^m error is 1.071% : see Fig. 4. For a random field with 10 method, the relative d = [10, 15, 20, 30, 40, 50], generalization errors of the ⁵ realizations random variables and $\beta = -2, -1, 0$, respectively. Again, the trained neural network are plotted in Fig. 5 for 0.5, our method still works with error smaller dimensionless magnitude of perturbation is about 10%, which is an acceptable tolerance in experiments. than

In 2D, a detailed dependence of EDL σ on random variables is given in Fig. 6 for $\beta = -2, -1, 0$, respectively. In these plots, we visualize EDL on the chosen random variable with

Figure 4: Generalization error of the trained neural network for random variables ranging over 13 [-5,5] with 1 photoluminescence datum in 2D. The relative L^{∞} error is 1.071%. The training data is visualized as a line and the testing data is visualized as scattering points.

all the other parameters fixed to be O. An interesting observation is that the larger the randomness is, the larger the EDL is. This is in contrast with experimental experiences and 3D results in Fig. 10. We attribute this observation to the particularity of dimension, which may deserve further experimental confirmations.

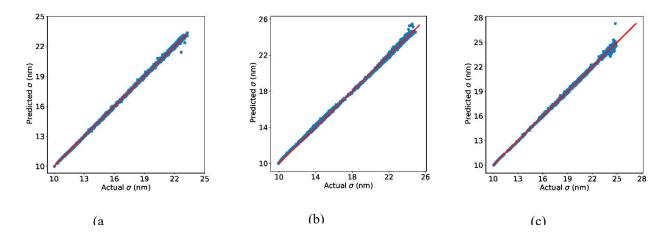


Figure 5: Generalization error of the trained neural network for random variables ranging over [-5, 5]with $\[Generalization error of the trained neural network for random variables ranging over <math>[-5, 5]$ with $\[Generalization error is 5.784\%$ when $\beta = -2$; (b) The relative L^{∞} error is 3.327\% when $\beta = -1$; (c) The relative L^{∞} error is 9.184% when $\beta = 0$. The training data is visualized as a line and the testing data is visualized as scattering points.

Figure 3: Generalization error of the trained neural network for random variables ranging over [-5, 5]with \mathfrak{E} photoluminescence data and $\beta = 0$ in 2D. (a) The relative L^{∞} error is 5.784% when $\beta = -2$; (b) The relative L^{∞} error is 3.327% when $\beta = -1$; (c) The relative L^{∞} error is 9.184%when $\beta = 0$. The training data is visualized as a line and the testing data is visualized as scattering points.

For the accuracy check in the 3D case, the reference PL data are generated using 5 realizations with out-of-plane thicknesses $d_i = 10, 15, 20, 25$ nm and $\sigma = 5$ nm in the absence of randomness. Afterwards, randomness is added with $K_1 = K_2 = 5$, i.e., $\theta(\omega_1)$

 $\theta(\omega_2)$ are arrays with ⁵ variables. QMC sampling is used to generate 20000 points and 15000 data are used as the with the corresponding EDL obtained by solving (9)-(12). The first training set, while the remaining data are used to check the predictability of the trained neural network; L[∞] errors of EDL are 0.270%, 0.368% 0.532% see Fig. 7. Relative and for $\beta = -2, -1, 0$, respectively. For completeness, we also plot the convergence history of the training process in terms of the iteration number in Fig. 8. It is known that the random field is closer to $\beta = 0$ and thus is more difficult to be trained. However, uniform the white noise when generalization errors for three different scenarios are observed, implying the robustness of trained neural network. Moreover, the size of training data set is small in the sense that only linear growth with respect to the dimension of random variables is observed, in contrast to other sampling techniques which either have the curse of dimensionality or low accuracy.

(\perp) Information extraction

The trained neural network fits a high-dimensional function for EDL in terms of surface roughness. Rich information can be extracted based on the fitted function. We demonstrate this using three examples.

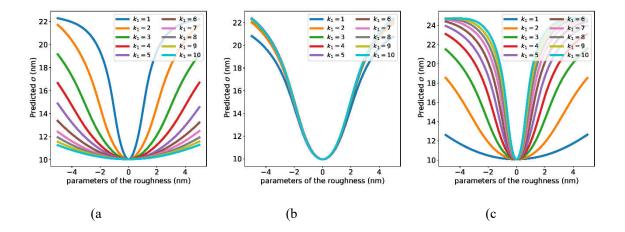


Figure 6 :Dependence of exciton diffusion length on random variables in 2D. (a) $\beta = -2$; (b) $\beta = -1$; (c) $\beta = 0$.

 $\beta = -2, -1, 0$. PL data are Expectations of EDL in 3D are recorded in Table 3 for $\beta = -2$, the EDL is ⁵ nm. When generated using the 1D model with the reference EDL 5 nm, which implies the equivalence between the 3D model and the 1D model. However, close to ⁵ nm. We attribute this difference to the $\beta = 0$, the EDL is clearly away from when modeling error between the 1D model and the 3D model with a surface roughness characterized by (9) $\beta = 0$. So far, the 1D model is largely used in the literature to extract the EDL^{[6][9][47][74]}. The with main assumption underlying the modeling is the high crystalline order of the organic material. When $\beta = -2$, long-range ordering exists in the random interface, which implicitly connects with the crystalline ordering of the material. Therefore, in this case, the 3D model and the 1D model are $\beta = 0$, only short-range ordering exists. As a consequence, the 3D equivalent. However, when model and the 1D model are not equivalent any more. This has been verified to be true over a range of EDLs. Given a surface roughness from the experimental measurement, we can fit a function of form (9) β using discrete Fourier transform, from which we can get the decay rate and thus decide whether the 1D model is adequate or not. It is worth mentioning that similar results are observed in 2D using the asymptotics-based approach^[50].

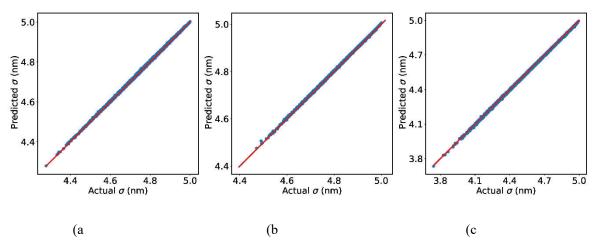


Figure 7: Uniform generalization error of the trained neural network for exciton diffusion length when $\beta = -2, -1, 0$. Relative L^{∞} errors of exciton diffusion length are 0.270%, 0.368% and 0.532%, respectively. The training data is visualized as a line and the testing data is visualized as scattering points.

Contour plots of the fitted EDL on random variables are given in Fig. 9 when $\beta = -2, 0$, respectively. In each subfigure, EDL σ is plotted as a function of $\theta_{k_1}(\omega_1)$ and $\theta_{k_2}(\omega_2)$, where $k_1, k_2 = 1, 2, 3, 4, 5$ and all the remaining random variables are set to be 0. A direct comparison between $\beta = -2$ and $\beta = 0$ in Fig. 9 illustrates the directional (anisotropic) dependence of EDL on random variables, due to different decay rates of random variables in the surface roughness. $\frac{2}{9}$ 1 $\frac{2}{9}$ $\frac{2}{9}$ $\frac{2}{9}$

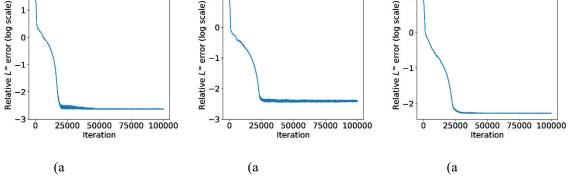
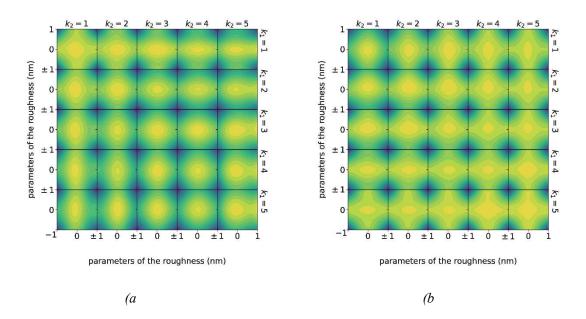


Figure 8: Relative L^{∞} error of exciton diffusion length in terms of iteration number for $\beta = -2, -1, 0$, respectively.

Fig. 10 provides a detailed demonstration of the dependence of EDL on random variables for $\beta = -2, -1, 0$. For illustration, we keep $\theta(\omega_2) = [1;0;0;0;0]$ fixed in the left column and $\theta(\omega_1) = [1;0;0;0;0]$ fixed in the right column. One distinct difference between 3D and 2D is that the maximum EDL is approached in the absence of randomness in 3D, in contrast to the minimum EDL in 2D. The 3D result is reasonable since experimentally larger EDL is observed if the effect of surface roughness is minimized, while the 2D result is also of interest due to the unique dimensional $\beta = -2$, the EDL is more sensitive to the lower-order modes (smaller kdependence. When $\beta = 0$, the trend is completely and is less sensitive to the high-order modes (larger ^{*k*}). When opposite. This observation provides a detailed connection between surface roughness and EDL, which also sheds light on the experimental design. Given a surface roughness characterized by (9), we have β , from which we know which mode is of the most importance. Consequently, the value of



targeted experimental techniques can be applied to improve the opto-electronic performance.

Figure 9: Contour plots of exciton diffusion length on random variables in 3D when $\beta = -2$, 0, respectively.

Conclusion

In summary, we have developed a novel method based on quasi-Monte Carlo sampling and ResNet to approximate the exciton diffusion length in terms of surface roughness parametrized by a high-dimensional random field. This method extracts a function for exciton diffusion length over the entire parameter space. Rich information, such as landscape profile and mode dependence, can be extracted with unprecedented details. Useful information regarding the modeling error and the experimental design can be provided, which sheds lights on how to reduce the modeling error and how to design better experiments to improve opto-electronic properties of organics materials.

Acknowledgments

L. Lyu acknowledges the financial support of Undergraduate Training Program for Innovation and

Entrepreneurship, Soochow University (Projection 201810285019Z). Z. Zhang acknowledges the financial support of Hong Kong RGC grants (Projects 27300616, 17300817, and 17300318) and National Natural Science Foundation of China via grant 11601457, Seed Funding Programme for Basic Research (HKU), and Basic Research Programme of The Science, Technology and Innovation Commission of Shenzhen Municipality (JCYJ20180307151603959). J. Chen acknowledges the financial support by National Natural Science Foundation of China via grants 21602149, 11971021, and National Key R&D Program of China (No. 2018YFB0204404). Part of the work was done when J. Chen was visiting Department of Mathematics, University of Hong Kong. J. Chen would like to thank its hospitality.

参考文献:

- J. D. Myers, J. Xue, Organic semiconductors and their applications in photovoltaic devices, Polymer Reviews 52 (2012) 1–37.
- Y.-W. Su, S.-C. Lan, K.-H. Wei, Organic photovoltaics, Materials Today 15 (2012) 554 – 562.
- [3] M. Pope, C. E. Swenberg, Electronic processes in organic crystals and polymers, Oxford university press, New York, 1999.
- [4] S. Forrest, The path to ubiquitous and low-cost organic electronic appliances on plastic, Nature 428 (2004) 911–918.
- [5] P. Peumans, A. Yakimov, S. Forrest, Small molecular weight organic thin-film photodetectors and solar cells, Journal of Applied Physics 93 (2003) 3693–3723.
- [6] J. Lin, O. Mikhnenko, J. Chen, Z. Masri, A. Ruseckas, A. Mikhailovsky, R. Raab, J. Liu, P. Blom, M. Loi, C. García-Cervera, I. Samuel, T.-Q. Nguyen, Systematic study of exciton diffusion length in organic semiconductors by six experimental methods, Materials Horizons 1 (2014) 280–285.

- [7] S. Sanaur, A. Whalley, B. Alameddine, M. Carnes, C. Nuckolls, Jet-printed electrodes and semiconducting oligomers for elaboration of organic thin-film transistors, Organic Electronics 7 (2006) 423–427.
- [8] A. R. Murphy, J. M. J. Fréchet, Organic semiconducting oligomers for use in thin film transistors, Chemical Reviews 107 (2007) 1066–1096.
- [9] L. Pettersson, L. Roman, O. Inganäs, Modeling photocurrent action spectra of photovoltaic devices based on organic thin films, Journal of Applied Physics 86 (1999) 487–496.
- [10] O. Mikhnenko, H. Azimi, M. Scharber, M. Morana, P. Blom, M. Loi, Exciton diffusion length in narrow bandgap polymers, Energy and Environmental Science 5 (2012) 6960–6965.
- [11] P. De Oliveira Neto, J. Teixeira, W. Da Cunha, R. Gargano, G. E Silva, Electron-lattice coupling in armchair graphene nanoribbons, Journal of Physical Chemistry Letters 3 (2012) 3039–3042.
- [12] C. Brabec, J. Hauch, P. Schilinsky, C. Waldauf, Production aspects of organic photovoltaics and their impact on the commercialization of devices, MRS Bulletin 30 (2005) 50–52.
- [13] J.-L. Brédas, J. Norton, J. Cornil, V. Coropceanu, Molecular understanding of organic solar cells: The challenges, Accounts of Chemical Research 42 (2009) 1691–1699.
- [14] Y. Terao, H. Sasabe, C. Adachi, Correlation of hole mobility, exciton diffusion length, and solar cell characteristics in phthalocyanine/fullerene organic solar cells, Applied Physics Letters 90 (2007) 103515.
- [15] S. Menke, W. Luhman, R. Holmes, Tailored exciton diffusion in organic photovoltaic cells for enhanced power conversion efficiency, Nature Materials 12 (2013) 152 - 157.
- [16] Antoniadis, Rothberg, Papadimitrakopoulos, W. Yan, Galvín, Abkowitz, Enhanced

carrier photogeneration by defects in conjugated polymers and its mechanism., Physical review. B, Condensed matter 50 20 (1994) 14911–14915.

- [17] T. Mullenbach, I. Curtin, T. Zhang, R. Holmes, Probing dark exciton diffusion using photovoltage, Nature Communications 8 (2017) 14215.
- [18] C. Tang, Two-layer organic photovoltaic cell, Applied Physics Letters 48 (1986) 183–185.
- [19] S.-B. Rim, R. Fink, J. Schöneboom, P. Erk, P. Peumans, Effect of molecular packing on the exciton diffusion length in organic solar cells, Applied Physics Letters 91 (2007) 173504.
- [20] D. Markov, C. Tanase, P. Blom, J. Wildeman, Simultaneous enhancement of charge transport and exciton diffusion in poly(p-phenylene vinylene) derivatives, Physical Review B - Condensed Matter and Materials Physics 72 (2005) 045217.
- [21] S. Scully, M. McGehee, Effects of optical interference and energy transfer on exciton diffusion length measurements in organic semiconductors, Journal of Applied Physics 100 (2006) 034907.
- [22] C. Goh, S. Scully, M. McGehee, Effects of molecular interface modification in hybrid organic-inorganic photovoltaic cells, Journal of Applied Physics 101 (2007) 114503.
- [23] P. Shaw, A. Ruseckas, I. Samuel, Exciton diffusion measurements in poly(3-hexylthiophene), Advanced Materials 20 (2008) 3516–3520.
- [24] Y. Wu, Y. Zhou, H. Wu, Y. Zhan, J. Zhou, S. Zhang, J. Zhao, Z. Wang, X. Ding, X. Hou, Metal-induced photoluminescence quenching of tri-(8-hydroxyquinoline) aluminum, Applied Physics Letters 87 (2005).
- [25] M. Theander, A. Yartsev, D. Zigmantas, V. Sundström, Photoluminescence quenching at a heterojunction, Physical Review B - Condensed Matter and Materials Physics 61 (2000) 12957–12963.

- [26] O. Mikhnenko, J. Lin, Y. Shu, J. Anthony, P. Blom, T.-Q. Nguyen, M. Loi, Effect of thermal annealing on exciton diffusion in a diketopyrrolopyrrole derivative, Physical Chemistry Chemical Physics 14 (2012) 14196–14201.
- [27] S. Cook, A. Furube, R. Katoh, L. Han, Estimate of singlet diffusion lengths in pcbm films by timeresolved emission studies, Chemical Physics Letters 478 (2009) 33–36.
- [28] A. Lewis, A. Ruseckas, O. Gaudin, G. Webster, P. Burn, I. Samuel, Singlet exciton diffusion in meh-ppv films studied by exciton-exciton annihilation, Organic Electronics 7 (2006) 452–456.
- [29] Z. Masri, A. Ruseckas, E. Emelianova, L. Wang, A. Bansal, A. Matheson, H. Lemke,
 M. Nielsen, H. Nguyen, O. Coulembier, P. Dubois, D. Beljonne, I. Samuel, Molecular
 weight dependence of exciton diffusion in poly(3-hexylthiophene), Advanced Energy
 Materials 3 (2013) 1445–1453.
- [30] P. Shaw, A. Ruseckas, J. Peet, G. Bazan, I. Samuel, Exciton exciton annihilation in mixed-phase polyfluorene films, Advanced Functional Materials 20 (2010) 155–161.
- [31] D. Qin, P. Gu, R. Dhar, S. Razavipour, D. Ban, Measuring the exciton diffusion length of c 60 in organic planar heterojunction solar cells, Physica Status Solidi (A) Applications and Materials Science 208 (2011) 1967–1971.
- [32] T. Stübinger, W. Brütting, Exciton diffusion and optical interference in organic donor-acceptor photovoltaic cells, Journal of Applied Physics 90 (2001) 3632–3641.
- [33] C. Yang, Z. Tang, W. Ge, J. Wang, Z. Zhang, X. Jian, Exciton diffusion in light-emitting organic thin films studied by photocurrent spectra, Applied Physics Letters 83 (2003) 1737–1739.
- [34] J. Halls, K. Pichler, R. Friend, S. Moratti, A. Holmes, Exciton diffusion and dissociation in a poly(pphenylenevinylene)/c60 heterojunction photovoltaic cell, Applied Physics Letters 68 (1996) 3120–3122.

- [35] A. Ghosh, T. Feng, Merocynanine organic solar cells, Journal of Applied Physics 49 (1978) 5982–5989.
- [36] J. Wagner, T. Fritz, H. Böttcher, Computer modelling of organic thin film solar cells. i. exciton model of photocurrent generation, Physica Status Solidi (a) 136 (1993) 423–432.
- [37] V. Bulović, S. Forrest, Excitons in crystalline thin films of 3,4,9,10-perylenetetracarboxylic dianhydride studied by photocurrent response, Chemical Physics Letters 238 (1995) 88–92.
- [38] S.-B. Rim, P. Peumans, The effects of optical interference on exciton diffusion length measurements using photocurrent spectroscopy, Journal of Applied Physics 103 (2008) 124515.
- [39] A. Huijser, T. Savenije, A. Shalav, L. Siebbeles, An experimental study on the molecular organization and exciton diffusion in a bilayer of a porphyrin and poly(3-hexylthiophene), Journal of Applied Physics 104 (2008) 034505.
- [40] M. Fravventura, J. Hwang, J. Suijkerbuijk, P. Erk, L. Siebbeles, T. Savenije, Determination of singlet exciton diffusion length in thin evaporated c 60 films for photovoltaics, Journal of Physical Chemistry Letters 3 (2012) 2367–2373.
- [41] J. Kroeze, T. Savenije, M. Vermeulen, J. Warman, Contactless determination of the photoconductivity action spectrum, exciton diffusion length, and charge separation efficiency in polythiophene-sensitized tio2 bilayers, Journal of Physical Chemistry B 107 (2003) 7696–7705.
- [42] R. Lunt, J. Benziger, S. Forrest, Relationship between crystalline order and exciton diffusion length in molecular organic semiconductors, Advanced Materials 22 (2010) 1233–1236.
- [43] K. Bergemann, S. Forrest, Measurement of exciton diffusion lengths in optically thin

organic films, Applied Physics Letters 99 (2011) 243303.

- [44] B. Rand, D. Cheyns, K. Vasseur, N. Giebink, S. Mothy, Y. Yi, V. Coropceanu, D. Beljonne, J. Cornil, J.-L. Brédas, J. Genoe, The impact of molecular orientation on the photovoltaic properties of a phthalocyanine/fullerene heterojunction, Advanced Functional Materials 22 (2012) 2987–2995.
- [45] R.Lunt, N.Giebink, A.Belak, J.Benziger, S.Forrest,

Excitondiffusionlengthsoforganicsemiconductor thin films measured by spectrally resolved photoluminescence quenching, Journal of Applied Physics 105 (2009) 053711.

- [46] T. Mullenbach, K. McGarry, W. Luhman, C. Douglas, R. Holmes, Connecting molecular structure and exciton diffusion length in rubrene derivatives, Advanced Materials 25 (2013) 3689–3693.
- [47] J. Chen, J. Lin, T.-Q. Nguyen, Towards a unified macroscopic description of exciton diffusion in organic semiconductors, Communications in Computational Physics 20 (2016) 754–772.
- [48] L. Hong, Z. Hu, G. Liu, Monte carlo methods for value-at-risk and conditional value-at-risk: A review, ACM Transactions on Modeling and Computer Simulation 24 (2014) 5.
- [49] D. Xiu, J. S. Hesthaven, High-order collocation methods for differential equations with random inputs, SIAM J. Scientific Computing 27 (2005) 1118–1139.
- [50] J. Chen, L. Lin, Z. Zhang, X. Zhou, Estimation of exciton diffusion lengths of organic semiconductors in random domains, Journal of Computational Physics 376 (2019) 894–912.
- [51] A. Oliynyk, L. Adutwum, B. Rudyk, H. Pisavadia, S. Lotfi, V. Hlukhyy, J. Harynuk, A. Mar, J. Brgoch, Disentangling structural confusion through machine learning: Structure

prediction and polymorphism of equiatomic ternary phases abc, Journal of the American Chemical Society 139 (2017) 17870–17881.

- [52] A. Oliynyk, L. Adutwum, J. Harynuk, A. Mar, Classifying crystal structures of binary compounds ab through cluster resolution feature selection and support vector machine analysis, Chemistry of Materials 28 (2016) 6672–6681.
- [53] G. Hautier, C. Fischer, A. Jain, T. Mueller, G. Ceder, Finding natures missing ternary oxide compounds using machine learning and density functional theory, Chemistry of Materials 22 (2010) 3762–3767.
- [54] O. Isayev, C. Oses, C. Toher, E. Gossett, S. Curtarolo, A. Tropsha, Universal fragment descriptors for predicting properties of inorganic crystals, Nature Communications 8 (2017) 15679.
- [55] T. Xie, J. Grossman, Crystal graph convolutional neural networks for an accurate and interpretable prediction of material properties, Physical Review Letters 120 (2018) 145301.
- [56] M. De Jong, W. Chen, R. Notestine, K. Persson, G. Ceder, A. Jain, M. Asta, A. Gamst,
 A statistical learningframeworkformaterialsscience:
 Applicationtoelasticmoduliofk-naryinorganicpolycrystalline compounds, Scientific
 Reports 6 (2016) 34256.
- [57] T.Wang, D.J.Wu, A.Coates, A.Y.Ng,

End-to-endtextrecognitionwithconvolutionalneuralnetworks, Proceedings of the 21st International Conference on Pattern Recognition (ICPR2012) (2012) 3304–3308.

[58] A. Krizhevsky, S. Ilya, H. G. E, ImageNet classification with deep convolutional neural networks, in: F. Pereira, C. J. C. Burges, L. Bottou, K. Q. Weinberger (Eds.), Advances in Neural Information Processing Systems 25, Curran Associates, Inc., 2012, pp. 1097–1105.

- [59] S. Ruhi, H. G. E., D. Anoop, Application of deep belief networks for natural language understanding, IEEE/ACM Trans. Audio, Speech and Lang. Proc. 22 (2014) 778–784.
- [60] C.Anitescu, E.Atroshchenko, N.Alajlan, T.Rabczuk,

Artificialneuralnetworkmethodsforthesolution of second order boundary value problems, Computers, Materials and Continua 59 (2019) 345–359.

- [61] H. Guo, X. Zhuang, T. Rabczuk, A deep collocation method for the bending analysis of kirchhoff plate, Computers, Materials & Continua 59 (2019) 433–456.
- [62] L. Yan, T. Zhou, An adaptive surrogate modeling based on deep neural networks for large-scale bayesian inverse problems, arXiv 1911.08926 (2020).
- [63] C. M. Ströfer, J. Wu, H. Xiao, E. Paterson, Data-driven, physics-based feature extraction from fluid flow fields using convolutional neural networks, Communications in Computational Physics 25 (2018) 625–650.
- [64] J. Han, L. Zhang, R. Car, W. E, Deep potential: A general representation of a many-body potential energy surface, Communications in Computational Physics 23 (2018) 629–639.
- [65] K. Li, K. Tang, T. Wu, Q. Liao, D3m: A deep domain decomposition method for partial differential equations, IEEE Access 8 (2020) 5283–5294.
- [66] Z.Wang, Z.Zhang,

Amesh-freemethodforinterfaceproblemsusingthedeeplearningapproach, Journal of Computational Physics 400 (2020) 108963.

- [67] H. Niederreiter, Random number generation and quasi-Monte Carlo methods, SIAM, 1992.
- [68] R. Caflisch, Monte Carlo and quasi-Monte Carlo methods, volume 7, Cambridge University Press, 1998.
- [69] J.Dick, F.Y.Kuo, I.H.Sloan, High-dimensionalintegration: thequasi-montecarloway,

ActaNumerica 22 (2013) 133–288.

- [70] K. He, X. Zhang, S. Ren, J. Sun, Deep residual learning for image recognition, CoRR 1512.03385 (2015).
- [71] G. F. Burkhard, E. T. Hoke, M. D. McGehee, Accounting for Interference, Scattering, and Electrode Absorption to Make Accurate Internal Quantum Efficiency Measurements in Organic and Other Thin Solar Cells, Advanced Materials 22 (2010) 3293–3297.
- [72] J. Chen, L. Lin, Z. Zhang, X. Zhou, Two-parameter asymptotic expansions for elliptic equations with small geometric perturbation and high contrast ratio, Asymptotic Analysis (2020). In press.
- [73] J. Nocedal, S. J. Wright, Numerical Optimization, Springer Series in Operations Research, SpringerVerlag, New York, 1999.
- [74] M. Guide, J. D. A. Lin, C. M. Proctor, J. Chen, C. Garcia-Cervera, T.-Q. Nguyen, Effect of copper metalation of tetrabenzoporphyrin donor material on organic solar cell performance, Journal of Material Chemistry A 2 (2014) 7890–7896.